



Carbon diffusion in supersaturated ferrite Insights from atomistic simulations

*[Bridging-Time Scale Techniques and their Applications in
Atomistic Computational Science]*

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Many thanks to the organisers!

C Diffusion in
supersaturated
Fe

Introduction

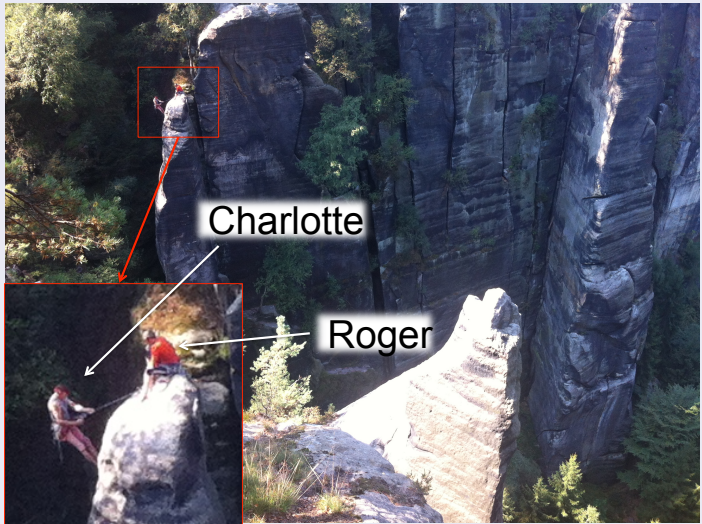
Methodology

Diffusion in
pure Fe

Diffusion in
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Conclusions

Simulation: a thrilling experience!



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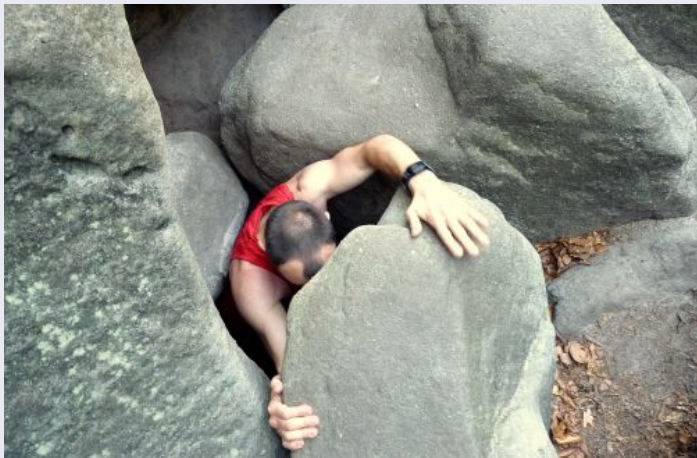
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Why does C diffuse so slowly in martensite ?

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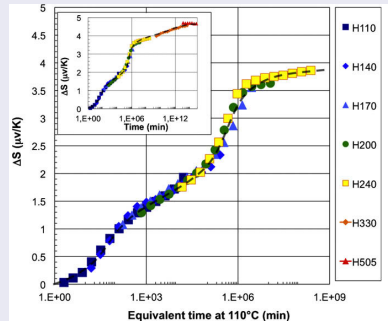
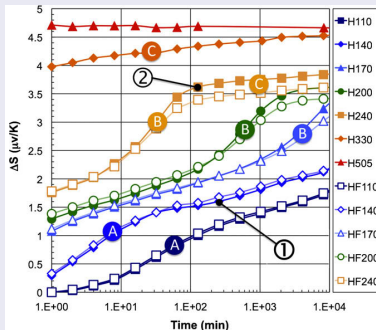
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Experiment: ageing of martensite

- Ageing monitored by thermoelectric power (*i.e.* resistivity)



- Time-temperature equivalency: $t = t_0 \exp \left[\frac{Q}{k_B T} \right]$
- $Q = 120 \text{ kJ/mol} = 1.25 \text{ eV}$ (0.85 eV for C in iron)

Why is the activation energy higher in martensite?

Why does C diffuse so slowly in martensite ?

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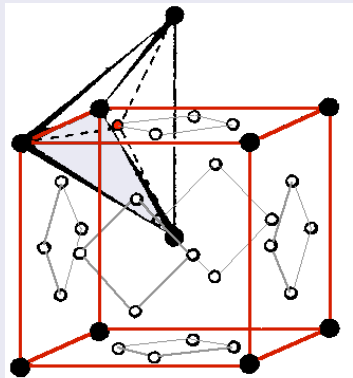
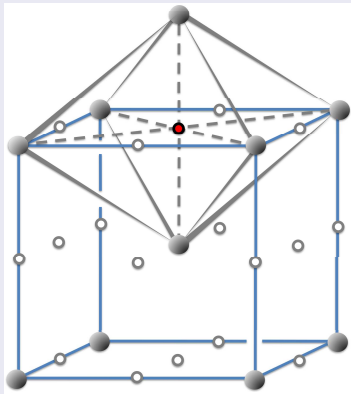
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Carbon diffusion: how?

- From octahedral to octahedral site
- Through tetrahedral site



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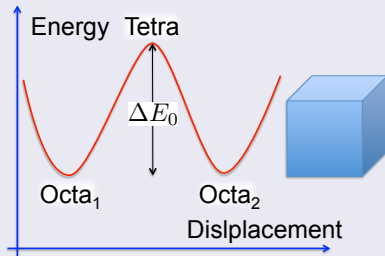
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The solution?

- 
[C. Zener, *Elast. Anelast. Metals*. Univ. Chigago Press (1948)]



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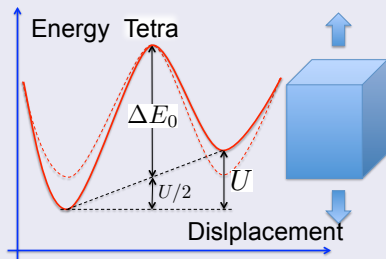
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The solution?

- 
 [C. Zener, *Elast. Anelast. Metals*. Univ. Chigago Press (1948)]



- 
 [M. Hillerts, *Acta Metall.* 7 (1959)]

approximately $\frac{1}{2}U$. As a consequence, the jump frequency will decrease by a factor $\exp(-NU/2RT)$, where $NU/2R = 1700c \text{ deg}$. Since long range diffusion is dominated by the highest energy barrier, the diffusion coefficient in martensite can be estimated as

$$D^{\text{martensite}} = D^{\text{ferrite}} \cdot \exp(-NU/2RT).$$

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- 1 Why does C diffuse so slowly in martensite ?
- 2 Methods, potential and systems
- 3 C diffusion in pure Fe
- 4 C diffusion in supersaturated Fe
- 5 Conclusions

Iron-carbon EAM potential

-  [Becquart *et al*, *Comp. Mat. Sc.* **40** (2007)]
- based on Mendeleev Fe potential
- fitted from C-Va and C-C interaction energies from DFT
- reproduce tetragonality of Fe-C martensite

Simulation box for MD

- 2000 Fe atoms and 1, 174 or 250 C atoms (small system)!
- Metropolis Monte-Carlo to relax the box
 - C is forced to remain ordered !
- MD simulation within the NVE ensembles at zero pressure
- 50 ns MD runs

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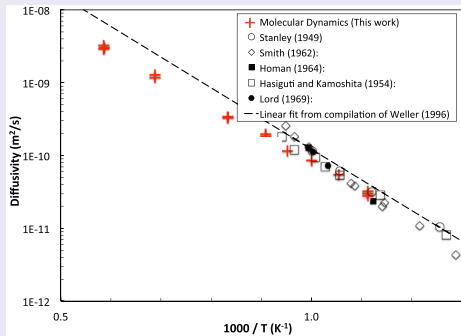
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MD results

- Diffusivity from Mean Square Displacements

$$\left\langle [\vec{r}_i(t_n) - \vec{r}_i(0)]^2 \right\rangle_{t,i} = \frac{\sum_i \sum_{j=1}^{n_{\max} - n} [\vec{r}_i((n+j)\delta t) - \vec{r}_i(j\delta t)]^2}{n_C(n_{\max} - n)} \quad D = \frac{\left\langle [\vec{r}_i(t_n) - \vec{r}_i(0)]^2 \right\rangle_{t,i}}{6t_n}$$



- Agreement with experiments (high temperature ?).

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Accounting for thermal expansion

- MD at various temperature under zero pressure

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$$\epsilon_{therm} = e_1 T + e_2 T^2$$

- Diffusivity equation

Accounting for thermal expansion

- MD at various temperature under zero pressure

$$\epsilon_{therm} = e_1 T + e_2 T^2$$

- Diffusivity equation

$$\frac{d \ln D}{d(1/T)} = -\frac{\Delta E(T)}{R}$$

How to determine $\Delta E(T)$?

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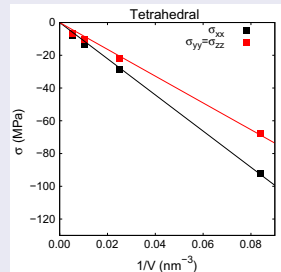
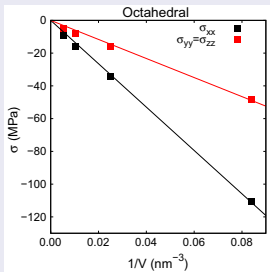
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Conclusions

A bit of elasticity theory: (1) dipole moment tensor

- One C atom in a stress free box \rightarrow dilatation
- One C atom in a fixed volume box \rightarrow stress field



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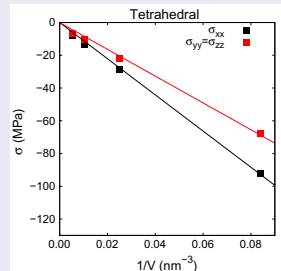
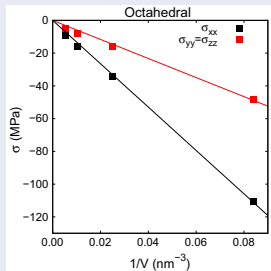
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$$\sigma_{ij}^o = \frac{1}{V} P_{ij}^o \quad P_{ij}^{o,t} = \begin{bmatrix} P_{\otimes}^{o,t} & 0 & 0 \\ 0 & P_{\bullet}^{o,t} & 0 \\ 0 & 0 & P_{\bullet}^{o,t} \end{bmatrix} \quad \sigma_{ij}^t = \frac{1}{V} P_{ij}^t$$

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A bit of elasticity theory: (2) interaction energy

- Interaction energy between a C atom and a far field strain:

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A bit of elasticity theory: (2) interaction energy

- Interaction energy between a C atom and a far field strain:

$$E_{inter} = P_{ij}\epsilon_{ij}$$

Back to thermal expansion

- Energy barrier variation with temperature:

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A bit of elasticity theory: (2) interaction energy

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- Energy barrier variation with temperature:

$$\begin{aligned}\Delta E(T) &= \Delta E_0 + (P_{ij}^t - P_{ij}^o) \mathbb{I}_{ij} \epsilon_{therm} \\ &= \Delta E_0 + \Delta P^{iso} \epsilon_{therm}\end{aligned}$$

- Diffusivity:

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$$\ln D = - \int \frac{\Delta E(T)}{R} d(1/T)$$

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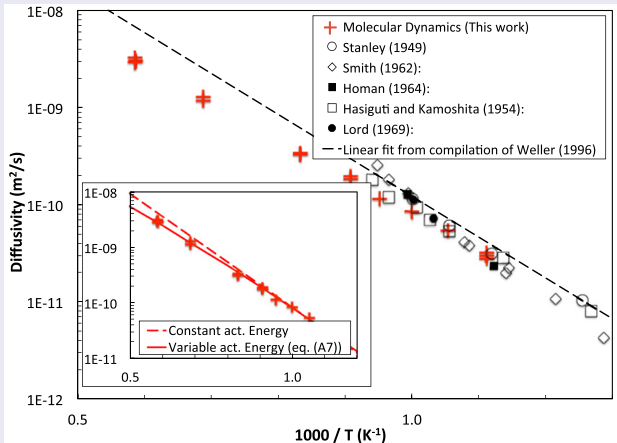
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MD results



- Not following experiments...
- ... but we understand MD non linearity!

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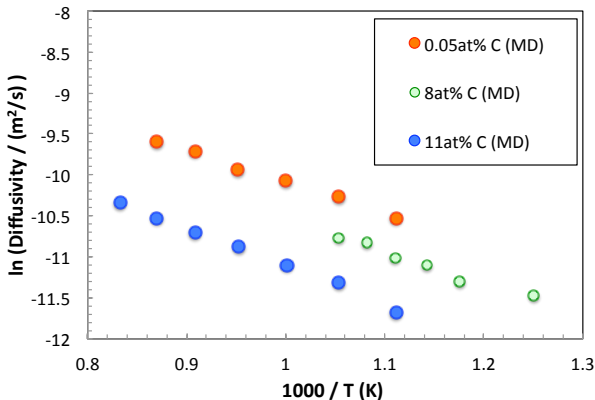
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- If D_0 is constant, why does activation energy varies ?

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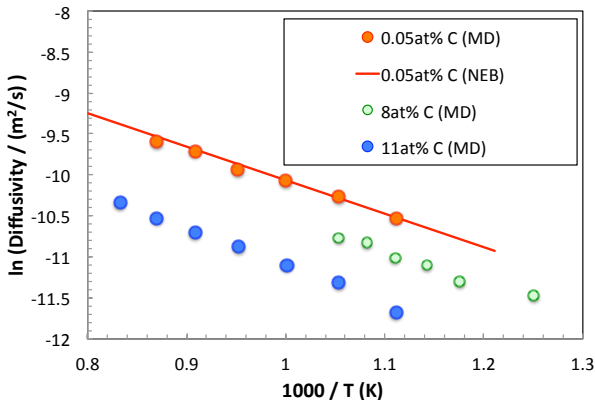
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- Nice fit, but only for low carbon concentration !

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Another bit of elasticity theory: (3) Effect of C concentration

- Interaction energy:

$$E_{inter} = P_{ij}\epsilon_{ij} \text{ with } P_{ij}^o = \begin{bmatrix} P_{\otimes}^o & 0 & 0 \\ 0 & P_{\bullet}^o & 0 \\ 0 & 0 & P_{\bullet}^o \end{bmatrix}$$

- Ordered C induced elastic distortion :

C diffusion in pure Fe

Another bit of elasticity theory: (3) Effect of C concentration

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- Ordered C induced elastic distortion :

$$\epsilon_{ij}^{nC} = \frac{n_C}{V} S_{ijkl} P_{kl}$$

- Hillert's idea:

Another bit of elasticity theory: (3) Effect of C concentration

- Interaction energy:

$$E_{inter} = P_{ij}\epsilon_{ij} \text{ with } P_{ij}^o = \begin{bmatrix} P_{\otimes}^o & 0 & 0 \\ 0 & P_{\bullet}^o & 0 \\ 0 & 0 & P_{\bullet}^o \end{bmatrix}$$

- Ordered C induced elastic distortion :

$$\epsilon_{ij}^{nC} = \frac{n_C}{V} S_{ijkl} P_{kl}$$

- Hillert's idea:

$$\begin{aligned} \Delta E &= \Delta E_0 + \frac{1}{2}(E^{unfav.} - E^{fav.}) \\ &= \Delta E_0 + \frac{1}{2}(E_{inter}^{o,unfav.} - E_{inter}^{o,fav.}) \\ &= \Delta E_0 + \frac{1}{2}(P_{ij}^{o,unfav.} - P_{ij}^{o,fav.}) \epsilon_{ij}^{nC} \end{aligned}$$

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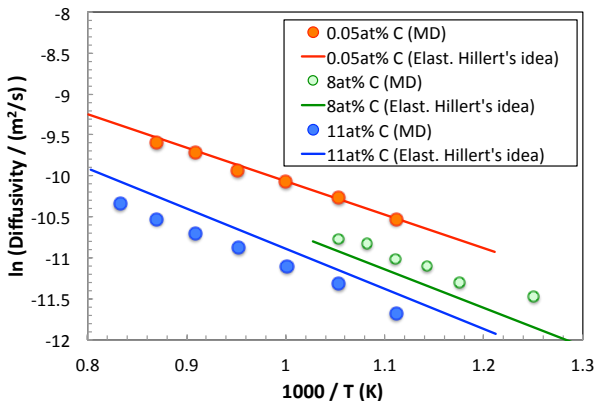
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MD results vs Elasticity theory following Hillert's idea



- Nice fit for low C concentration only !?!

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A las bit of elasticity theory: (4) accounting for octa and tetra

- Interaction energy:

$$E_{inter} = P_{ij}\epsilon_{ij} \text{ with } P_{ij}^{o,t} = \begin{bmatrix} P_{\otimes}^{o,t} & 0 & 0 \\ 0 & P_{\bullet}^{o,t} & 0 \\ 0 & 0 & P_{\bullet}^{o,t} \end{bmatrix}$$

- C induced elastic distortion:

$$\epsilon_{ij}^{nC} = \frac{n_C}{V} S_{ijkl} P_{kl}$$

- Energy barrier:

C diffusion in pure Fe

A las bit of elasticity theory: (4) accounting for octa and tetra

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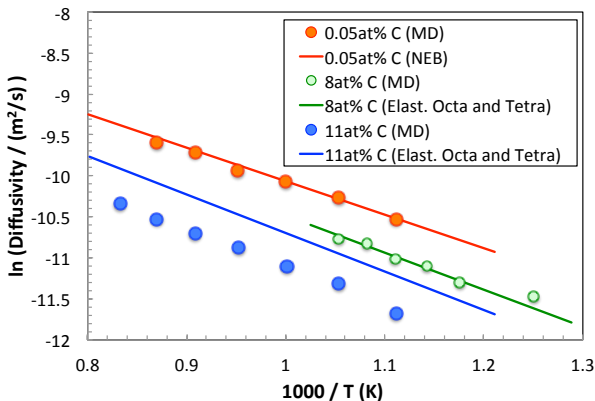
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MD results vs Elasticity theory accounting for Tetra and Octa



- Still not working for 11% !?!

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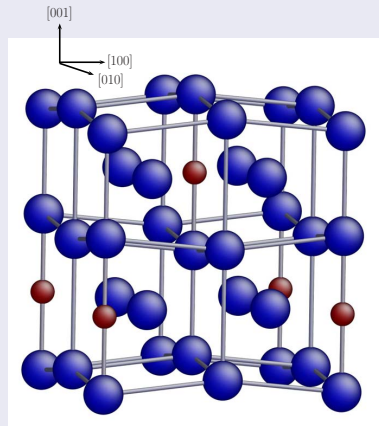
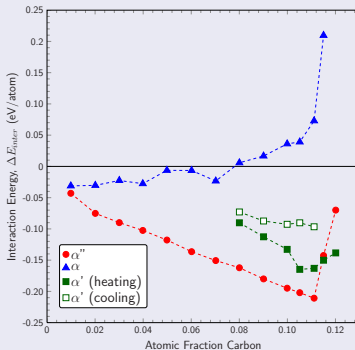
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What happens at 11%C?

- A new phase !?!



[Sinclair *et al*, *Phys. Rev. B*. **81** (2010)]

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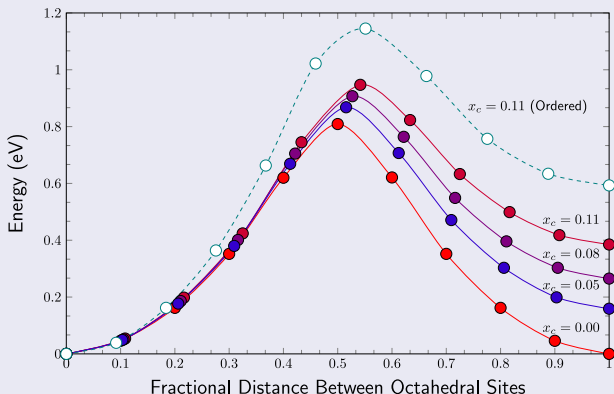
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How can it change diffusivity?

- NEB on strained simulations boxes:



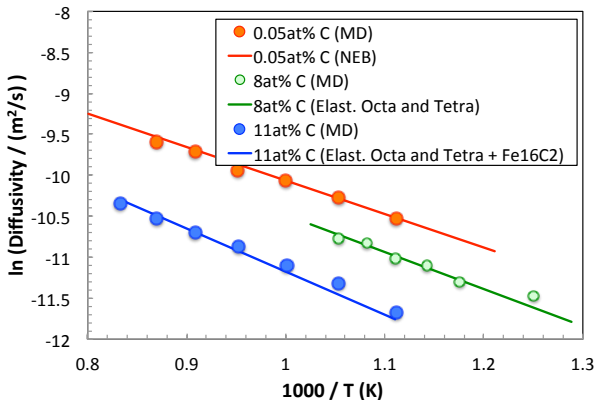
- Fe_{16}C_2 exhibits higher energy barrier

A mixture of fully ordered Fe_{16}C_2 and ordered 11%C?

- Order parameter η
 - $\eta = 1$ for Fe_{16}C_2
 - $\eta = 0$ for “random” ordered structure

$$\Delta E = k_B \ln \left(\eta \exp \left[\frac{\Delta E^{\text{Fe}_{16}\text{C}_2}}{k_b T} \right] + (1 - \eta) \exp \left[\frac{\Delta E^{\text{Fe}11\%C}}{k_b T} \right] \right)$$

Final comparison



- $D_0 = 1.1 \times 10^{-6} \text{ m}^2/\text{s}$ in all cases

Conclusion

- Ordered C in solid solution modifies diffusivity of C
- Hillert was (almost) right !
- Better to account for octa and tetra variation with C content



[Lawrence *et al*, *Modelling Simul. Mater. Sci. Eng.* 22 (2014)]

Outlook

- Diffusion of C within a Cottrell atmosphere
 - see Charlotte's talk on thursday
- Formation of carbides...